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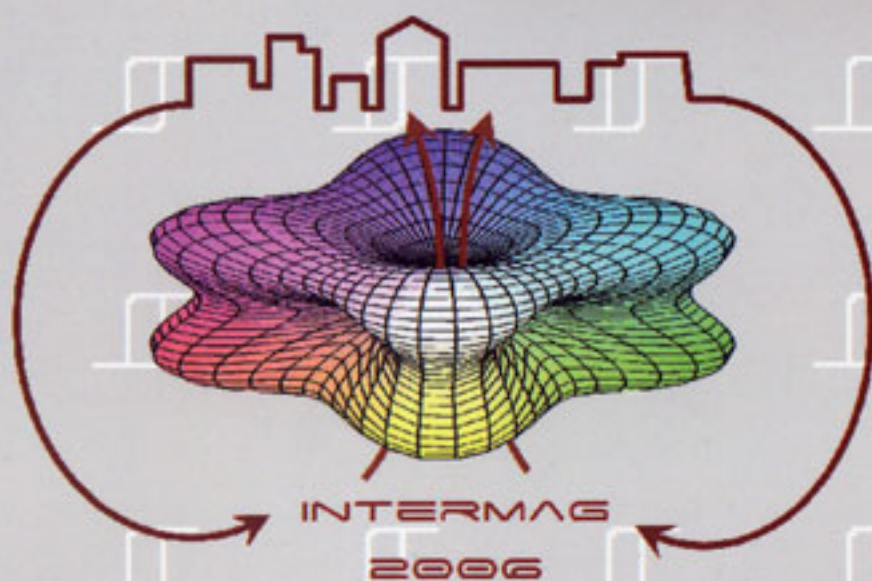
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## HW-06

Synthesis and Magnetic Properties of  $\text{LiFeO}_2$  Powders by a Sol-Gel Method.

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## Introduction

The commercial lithium battery with high energy density and good cycle life has been studied as a power source for portable electronics. Many research groups have investigated various cathode materials for the lithium secondary batteries such as a layered oxide;  $\text{LiMO}_2$  ( $M=\text{Co, Ni, Mn, Fe}$ ), which consists of alternating layers of trigonally distorted  $\text{MO}_6$  and  $\text{LiO}_6$  octahedral sharing edges[1]. Lithium iron oxides such as  $\text{LiFeO}_2$  and  $\text{LiFe}_5\text{O}_8$  are increasing scientific interest and are also promising candidates for cathode materials in rechargeable lithium batteries as well as low-cost substitutes to garnet materials ( $\text{Y}_3\text{Fe}_5\text{O}_{12}$ ) in microwave frequency applications[2].

## Experiments

Ordered  $\text{LiFeO}_2$  powders were prepared by the sol-gel method. Lithium acetate [ $\text{CH}_3\text{CO}_2\text{Li}$ ] and iron nitrate nonahydrate [ $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ ] were dissolved in ethanol and distilled water.  $\text{LiFeO}_2$  powders were annealed in the range of 300–1000 °C in air atmosphere for 6 h. The crystal structures were measured by X-ray diffraction (XRD) using  $\text{CuK}\alpha$  radiation. Thermal analysis such as thermogravimetry analysis (TGA) and differential thermal analysis (DTA) were performed on the dried powder obtained from the  $\text{LiFeO}_2$  precursor solution. The particle morphology of the samples was observed using a scanning electron microscope (SEM). Magnetic properties were measured using a vibrating sample magnetometer (VSM). Mössbauer spectra were recorded using a 40 mCi  $^{57}\text{Co}$  source in a Rh matrix with the spectrometer working at constant acceleration.

## Results and Discussion

DTA analysis indicated an exothermic peak. The peak at 245 °C is associated with the organic weight losses. However, the final peak appears at about 550 °C, indicating the complete thermal decomposition and crystallization of  $\alpha\text{-LiFeO}_2$  occur simultaneously.  $\alpha\text{-LiFeO}_2$  powders that were annealed at and above 600 °C have a single-phase. The x-ray diffraction patterns were refined by Rietveld refinement using the FULLPROF program. The crystal structure of the  $\alpha\text{-LiFeO}_2$  at room temperature, was determined to be cubic of  $Fm\bar{3}m$  space group with its lattice constant  $a = 4.1610 \text{ \AA} \pm 0.0005$ . The Bragg factors  $R_B$  and  $R_F$  were 5.56 % and 3.79 %, respectively. The SEM micrographs indicate the distribution of grains uniform with really spherical shape. The powders present average particle sizes of 110 nm. Mössbauer spectra of the  $\alpha\text{-LiFeO}_2$  were taken at various absorber temperatures from 4.2 to 295 K. The Néel temperature of  $\alpha\text{-LiFeO}_2$  is found to  $T_N = 90 \pm 2$  K. Figure 1 illustrates some of the spectra of the sample, which were composed of two sixline hyperfine patterns. The spectra for the samples at 4.2 K exhibit general sextets shape indicating ferromagnetic behaviors. The spectrum was fitted using the two magnetic components of hyperfine fields  $H_{\text{hf}} = 506$  and 478 kOe, isomer shifts  $\delta = 0.37$  and 0.36 mm/s corresponding to  $\text{Fe}^{3+}$  ions. Mössbauer spectra of  $^{57}\text{Fe}$  at room temperature for  $\alpha\text{-LiFeO}_2$  powders are shown paramagnetic behavior as demonstrated by the single quadrupole doublet with zero hyperfine fields. The hyper-

fine parameters for the sample are isomer shift  $\delta = 0.24$  mm/s and quadrupole splitting  $E_Q = 0.61$  mm/s, respectively. Figure 2 shows the fractional change of the average magnetic hyperfine field,  $[H_{\text{hf}}(T) - H_{\text{hf}}(0)] / H_{\text{hf}}(0)$ , as a function of reduced temperature  $T/T_N$ . The average magnetic hyperfine field decreases with increasing temperature according to  $[H_{\text{hf}}(T) - H_{\text{hf}}(0)] / H_{\text{hf}}(0) = -0.36(T/T_N)^{3/2} - 0.27(T/T_N)^{5/2}$  for  $T/T_N < 0.7$ , indicative of spin-wave excitation.  $\ln F$  versus  $T_2$  for  $\alpha\text{-LiFeO}_2$ , where  $F$  stands for the total resonance absorption to the recoil-free fraction  $f$ . From the temperature variation of the total Mössbauer absorption areas, we obtained the Debye temperatures of  $\alpha\text{-LiFeO}_2$  to be  $\Theta = 253 \pm 5$  K. The temperature dependence of the inverse molar susceptibilities was linear above 200 K for  $\alpha\text{-LiFeO}_2$ . The calculated Curie-Weiss temperature ( $\theta_p$ ) value was  $-160 \pm 5$  K for  $\alpha\text{-LiFeO}_2$ .

[1] Y. S. Lee, C. S. Yoon, Y. K. Sun, Y. Sato, Electrochemistry Communications, **4**, 727(2002).

[2] K. U. Kang, H. N. Oak, and C. S. Kim, J. Appl. Phys. **97**, 10F102(2005).

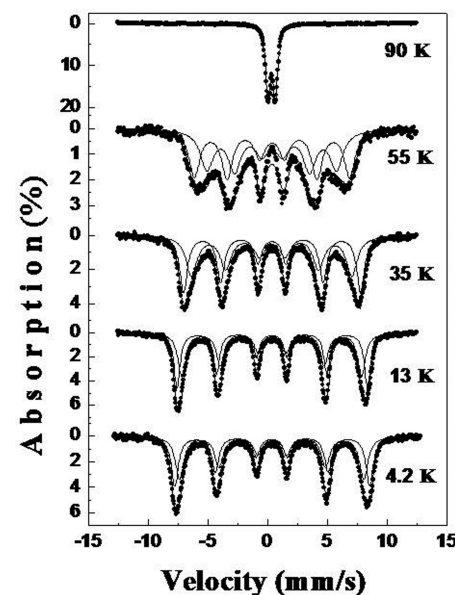


Fig. 1. Mössbauer spectra of  $\alpha\text{-LiFeO}_2$  at various temperatures.

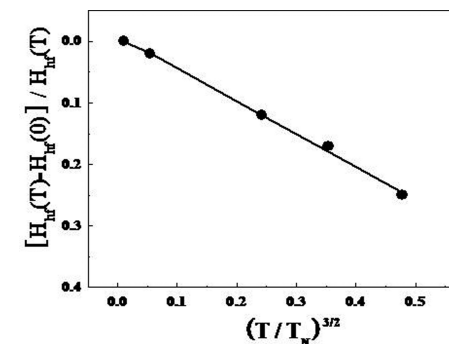


Fig. 2. Fractional change of the average magnetic hyperfine field,  $H_{\text{hf}}$ , as a function of  $(T/T_N)^{3/2}$  for  $\alpha\text{-LiFeO}_2$ .