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Applied-field dependent hyperfine structure and magnetic properties of Ni $_{0.8-}$ $_{x}$ Cu $_{x}$ Zn $_{0.7}$ Fe $_{2}$ O $_{4}$

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Introduction

Nickel-copper-zinc ferrites are interesting magnetic materials for multilayer chip inductor (MLCI) applications, which is due to the fact that these ferrites have high electrical resistivity, permeability, and low sintering temperature [1, 2]. Moreover those, Nickel-copper-zinc ferrites have interesting physical properties as a non-collinear magnetic structure. One feature making an investigation of the Ni-Cu-Zn ferrites especially noteworthy is the difference in cation distributions. Either the Zn^{2+} ions occupy only A sites, having a normal spinel structure, or Ni^{2+} , Cu^{2+} ions are only at B sites, forming the inverse spinel structure.

In this paper, we investigated applied-field dependent magnetic hyperfine structure of Ni²⁺, Cu²⁺, and Zn²⁺ ions for Ni_{0.8-x}Cu_xZn_{0.2}Fe₂O₄ as well as corresponding magnetic spin structure with applied-field Mö ssbauer spectroscopy.

Experiments

Single phased of polycrystalline $Ni_{0.8-x}Cu_xZn_{0.2}Fe_2O_4(x=0.2,\ 0.4,\ 0.6)$ was synthesized by the solid-state reaction method. NiO (99.99%), ZnO (99.99%), CuO (99.99%) and α -Fe₂O₃ powder (99.995%) were starting materials. These were mixed and annealed at 1000 °C for 24h in air. The identification of the crystal structure and determination of lattice constant were carried out by x-ray diffraction (XRD). The microstructure and morphology of the prepared sample were examined by a field emission scanning electron microscope (FE-SEM). Magnetic properties were measured by a vibrating sample magnetometer (VSM) and applied-field Mö ssbauer spectrometer, using a 57 Co (Rh) source in a constant acceleration mode. In order to separate sublattice lines, Mö ssbauer spectra were taken in the presence of an external magnetic field under 5 T parallel to the γ -ray direction at 4.2 K.

Results and discussion

The crystal structure of $Ni_{0.8}$, $Cu_{\star}Zn_{0.2}Fe_{2}O_{4}(x = 0.2, 0.4, 0.6)$ was determined by Rietveld FULL-PROF program. The Zn^{2+} and iron ions occupy A sites, and Ni^{2+} , Cu^{2+} , iron ions occupy B sites, forming $(Zn_0, Fe_0)^A[Ni_0, Cu, Fe_1]^BO_4(x = 0.2, 0.4, 0.6)$. The crystal structure was determined to be a cubic spinel with space group Fd-3m. The lattice constant a_0 increases from 8.373 to 8.373 Å with increasing Cu concentration. The magnetic hysteresis loop of Ni_{0.8.x}Cu_xZn_{0.2}Fe₂O₄ powders at room temperature with various Cu concentrations. The saturation magnetization (Ms) and coercivity (Hc) under the applied-field of 10 kOe, were 57.5 emu/g and 38 Oe for x = 0.2 sample, and 52.8 emu/g and 13 Oe for x = 0.6 sample, respectively, as shown Fig. 1. FE-SEM image of the Ni_{0.0} $_{\rm u}$ Cu $_{\rm u}$ Zn $_{\rm o}$ Fe $_{\rm o}$ O $_{\rm d}$ (x = 0.2, 0.4, 0.6) for morphology of the synthesized particles shows irregular shape and an inhomogeneous microstructure with discontinuous gain growth. According to the probability of distribution, we have analyzed Mö ssbauer spectra as 5 sets with six-lines at 4.2 K. Appliedfield Mö ssbauer spectra of the $Ni_{0.8-x}Cu_xZn_{0.2}Fe_xO_4(x = 0.2, 0.4, 0.6)$ were measured with parallel to the γ-ray direction under 5 T at 4.2 K, as shown Fig. 2. Hyperfine fields of A and B sites at 5 T was $H_{k}(A) = 521$ kOe, average $\langle H_{k}(B) \rangle = 447$ kOe for x = 0.2 sample, and $H_{k}(A) = 510$ kOe, $\langle H_{\rm h}(B) \rangle = 440$ kOe for x = 0.6 sample, respectively. Also, the second and fifth absorption lines of Mö ssbauer spectra were completely disappeared above 3 T. It means that the spins of Fe ions at A

and B sites are collinear to the external field. The Fe valence states were determined to be ferric with the isomer shift values.

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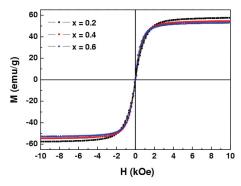


Fig. 1 The magnetic hysteresis loop of $Ni_{0.8-}$ $_xCu_xZn_{0.2}Fe_2O_4(x=0.2,\ 0.4,\ 0.6)$ at room temperature.

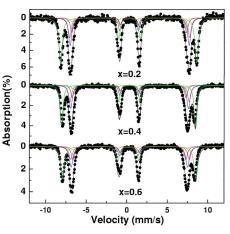


Fig.2 Mössbauer spectra of $Ni_{0.8-}$ $_xCu_xZn_{0.2}Fe_2O_4(x=0.2, 0.4, 0.6)$ at 4.2 K with 5T applied-magnetic field.